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# Enhanced Visible-Photocatalytic Activity of Anodic TiO<sub>2</sub> Nanotubes Film via Decoration with CuInSe<sub>2</sub> Nanocrystals

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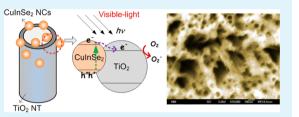
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**Supporting Information** 

**ABSTRACT:** The fabrication and photocatalytic properties of visiblelight driven  $CuInSe_2/TiO_2$  heterojunction films are reported.  $CuInSe_2$ nanoparticles (NCs) were synthesized using a solvothermal method and then decorated onto self-organized anodic  $TiO_2$  nanotube (NT) arrays through an electrophoretic deposition process, forming a  $CuInSe_2 NC/TiO_2 NT$  hetero-structure film. An increase in deposition time produced an increased amount of  $CuInSe_2 NCs$  loaded onto the  $TiO_2 NT$  arrays, expanding the light-absorption range of the CuInSe\_2



NCs/TiO<sub>2</sub> NTs film from 400 nm to 700 nm. Photocatalytic degradation results show that activities of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films were significantly enhanced compared to that of pure TiO<sub>2</sub> NTs (degradation rate constant *k* increased from  $3 \times 10^{-3}$  min<sup>-1</sup> to >1 × 10<sup>-2</sup> min<sup>-1</sup>). Particularly, the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs with 50 min electrophoretic deposition show the highest degradation rate,  $k = 1.6 \times 10^{-2}$  min<sup>-1</sup> (more than 5 times greater than that of the pure TiO<sub>2</sub> NTs film), due to optimization of CuInSe<sub>2</sub> NCs loading and a well-maintained open TiO<sub>2</sub> tube-mouth configuration.

KEYWORDS: TiO<sub>2</sub> nanotubes, CuInSe<sub>2</sub> nanocrystals, heterojunctions, photocatalytic activity

# 1. INTRODUCTION

During the past decades, organic pollution in water has been one of the most serious environmental problems, and great efforts have been applied using various technologies to eliminate these toxic chemicals from water.<sup>1-4</sup> Among these strategies, semiconductor photocatalysis has been shown to be a promising remediation technology with unique advantages, such as energy efficiency and no secondary pollution.<sup>5-7</sup> In particular, titanium dioxide (TiO<sub>2</sub>), a wide-band semiconductor  $(\sim 3.2 \text{ eV})$ , is particularly useful for organic degradation. When TiO<sub>2</sub> was properly irradiated, electron/hole pairs or excitons could be generated by incoming photons, which either migrate to the material surface or recombine and dissipate the energy as heat. Those on the surface can then participate in redox reactions and generate reactive oxygen species (ROS). Both the photo-generated electron/hole pairs and ROS contribute to the degradation and removal of organic pollutants from water.<sup>8,9</sup> TiO<sub>2</sub>-based materials are still among the most widely used semiconductor photocatalysts, due to their low cost, high catalytic activity, good chemical and mechanical stability, and non-toxicity to environment and human beings.<sup>10-14</sup>

However, there are two major concerns during the application of  $TiO_2$  photocatalysts: The first is how to recycle the  $TiO_2$  photocatalysts; most photocatalyst materials are used in powder form, which is very difficult to recover for beneficial

reuse. The second is how to increase the photocatalytic light absorption range. Due to its wide band-gap nature (~3.2 eV),  $TiO_2$  is excited only by ultraviolet light.<sup>15–18</sup> Separating and recycling suspended TiO<sub>2</sub> powder photocatalysts from treated water has many difficulties and energy cost problems. Immobilized TiO<sub>2</sub> photocatalyst materials have been shown to be good candidates to solve this problem and have attracted increased interest. One class of outstanding photocatalyst materials are anodic TiO<sub>2</sub> nanotube (NT) films, not only due to their large surface area but also due to their inherent immobilization, which can reduce the energy and material cost of separating the photocatalysts.<sup>19–23</sup> Nevertheless, their large band-gap along with a fast recombination rate of the photogenerated electron/hole pairs still limit their photocatalytic performance and hinder commercial and industrial applications. Many efforts have been attempted to further overcome such impediments, one of which is to build heterogeneous  $TiO_2$  photocatalysts with narrow band gaps.<sup>24,25</sup> For instance, TiO<sub>2</sub>/CdS or TiO<sub>2</sub>/CdSe heterojunction photocatalysts have been widely studied for organic pollutant decomposition under visible irradiation.<sup>26–30</sup> Here,

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CdS or CdSe, having a narrow band gap, not only acts as a visible-light sensitizer but also enhances the electron/hole pair separation in the heterojunction system at the same time. However, considering environmental regulations, the intrinsic toxicity of Cd casts doubts on the future applicability of the Cd compounds, despite their great electrical and optical properties. In light of this, Cd-free alternative materials were proposed, such as I-III-VI<sub>2</sub> nanocrystals with fewer environmental impacts than Cd,  $^{31-33}$  although so far synthesis of I-III-VI<sub>2</sub> nanocrystals is much less developed.

Herein, we report a new study on the fabrication and photocatalytic performance of a visible-light heterojunction photocatalyst, which consists of  $CuInSe_2$  nanocrystal (NCs)/TiO<sub>2</sub> NT arrays as shown in Figure 1a. CuInSe<sub>2</sub>, a direct energy

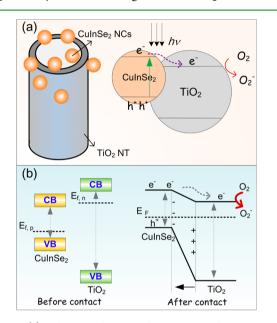


Figure 1. (a) Schematic diagram of CuInSe<sub>2</sub> NC/TiO<sub>2</sub> NT arrays heterojunction system. (b) Diagram of the band energy of CuInSe<sub>2</sub> and TiO<sub>2</sub> before and after contact, along with electron-hole pair separation and transmission CuInSe<sub>2</sub> /TiO<sub>2</sub> heterojunction system.

gap material (~1.04 eV), belongs to one of the I-III-VI<sub>2</sub> ternary compounds, has a large absorption coefficient, and is composed of much less toxic elements.<sup>33,34</sup> Meanwhile, the optical absorption of the CuInSe2 NCs lies within the visible-light range and therefore matches well to the solar spectrum. In this study, the CuInSe<sub>2</sub> NCs with uniform size distribution were synthesized using solvothermal methods and decorated onto highly ordered anodic TiO<sub>2</sub> NT arrays by using electrophoretic deposition (EPD) and subsequently used as a visible-light sensitizer. The as-obtained films have a two-component structure as CuInSe2 NCs/TiO2 NTs, with various CuInSe2 loading ratio. The CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT films were characterized by scanning electron microscopy, X-ray diffraction, UV-vis absorption spectroscopy, etc. Photocatalytic activities of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT heterojunction films were investigated by evaluating the photocatalytic degradation of methyl orange aqueous solution at room temperature.

#### 2. EXPERIMENTAL SECTION

All chemicals used in our experiment were received from commercial companies and used without further purification (analytical- grade reagents). Deionized water with a resistance of 18.3 M $\Omega$ ·cm was used in the experiment.

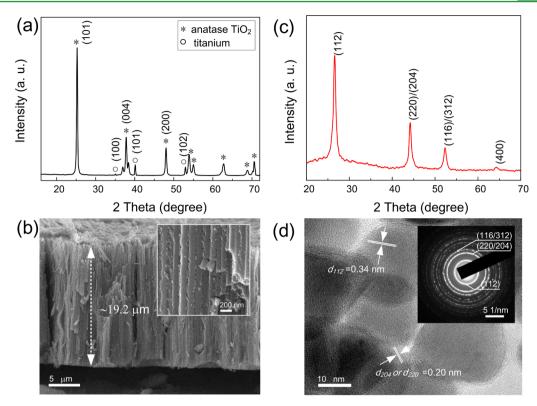
2.1. Synthesis of TiO<sub>2</sub> NT Arrays. TiO<sub>2</sub> nanotube arrays were synthesized by an electrochemical process in a fluorine-containing nonaqueous electrolyte as described by Grimes et al.<sup>35–37</sup> Briefly, Ti metal sheets (>99.6%, Beijing Nonferrous Metal Research Institute) were cut into small pieces  $(1.5 \text{ cm} \times 5 \text{ cm})$ , cleaned in sonication baths in acetone and ethanol each for 10 min, and then dried at room temperature. One piece of cleaned Ti sheet was taken as the anode with a platinum (Pt) foil as the counter electrode and directly immersed into the electrolyte (294 mL glycol, 6 mL H\_2O, and 1 g NH4F). A constant potential of 60 V was applied across the two electrodes with a distance of 2.5 cm by using a DC power supply (Model GPS-3303c, GW Instrument Co., Ltd Taiwan). The anodic oxidation was carried out in this two-electrode configuration for 3 h at 7 °C, obtaining highly ordered TiO<sub>2</sub> NT arrays on the Ti sheet substrate. To induce crystallinity, the highly ordered TiO<sub>2</sub> NT arrays were annealed at 450 °C for 2 h.

**2.2. Synthesis of CulnSe<sub>2</sub> NCs.** CuInSe<sub>2</sub> NCs were synthesized in a non-coordinating oleylamine solvent.<sup>38,39</sup> In a typical process, 1 mmol CuCl, 1 mmol InCl<sub>3</sub>·4H<sub>2</sub>O, 2 mmol Se powder, and 10 mL oleylamine were mixed together and loaded in a three-neck flask. The system was heated to 70 °C under a nitrogen atmosphere with stirring for 90 min to remove water and oxygen. Then the mixture solution was quickly heated to 220 °C and incubated for 4 h. After that, the reaction vessel was allowed to naturally cool to room temperature. Then well-dispersed CuInSe<sub>2</sub> NC black "inks" were obtained by further purification (using ethanol and chloroform) along with centrifugation. It is should be noted that, to obtain high quality CuInSe<sub>2</sub> NCs, vigorous stirring is required through the synthesis process. The typical product yield of the well-dispersed CuInSe<sub>2</sub> NCs was around 15%.

**2.3.** Decoration of TiO<sub>2</sub> NT Arrays with CulnSe<sub>2</sub> NCs by Using an Electrophoresis Method. Electrophoresis allows for simple fabrication of large-scale and complex-shaped thin films, with a high level of coating thickness control, with simple equipment required and low cost.<sup>40</sup> Therefore, we used this method to decorate the TiO<sub>2</sub> NT arrays with the CulnSe<sub>2</sub> NCs. First, the as-prepared CulnSe<sub>2</sub> NCs were re-dispersed in chloroform with a typical concentration of 10<sup>6</sup> mol L<sup>-1</sup>. Then, the highly-ordered TiO<sub>2</sub> NT arrays electrodes were immersed in the CulnSe<sub>2</sub> NCs solution, and a constant voltage of 150 V was applied for 25, 50, or 75 min. After the electrophoresis process, the electrodes were rinsed with chloroform several times to wash off the unbounded CulnSe<sub>2</sub> NCs. Finally, the CulnSe<sub>2</sub> NC-decorated TiO<sub>2</sub> NT arrays were annealed at 300 °C for 15 min to form CulnSe<sub>2</sub>/TiO<sub>2</sub> heterojunction photocatalyst.

**2.4. Characterization.** X-ray diffraction (XRD) analysis was employed to characterize the crystallinity of the as-synthesized  $TiO_2$  NT arrays and CuInSe<sub>2</sub> NCs (a D/max 2400 X Series X-ray diffractometer). The X-ray radiation source was Cu K $\alpha$ , obtained at 40 kV, 100 mA, and the scanning speed was 10°/min at a step of 0.02°. Transmission electron microscopy (TEM, JEM2100, JEOL Inc., Japan) was used to characterize the structural and morphological properties of the CuInSe<sub>2</sub> NCs. A field emission scanning electron microscopy (FESEM, JSM-7000F, JEOL Inc. Japan) was employed to characterize the morphology of the CuInSe<sub>2</sub>/TiO<sub>2</sub> films. UV–vis absorption spectra of the samples were obtained by a JASCO V-570 UV/VIS/NIR spectrometer. The specific surface area was obtained by scraping the TiO<sub>2</sub> NT arrays from the Ti substrate and then measuring the loose NTs by Brunauer-Emmett-Teller (BET) analysis via nitrogen adsorption at 77 K (model ASAP 2020, Micromeritics, USA).

**2.5. Photocatalytic activity measurement.** Photocatalytic activities of  $CuInSe_2/TiO_2$  heterojunction photocatalysts were evaluated on the basis of the degradation of methyl orange (MO) as model organic pollutants in aqueous solutions (40 mL total volume, 20 mM). The  $CuInSe_2/TiO_2$  NT films (3.0 × 1.5 cm<sup>2</sup> surface area, total mass = 0.04 g) were immersed in the solution and irradiated with six 4-W visible bulbs (Toshiba, Cool white, FL4W, Japan). Then the solution was magnetically stirred in the dark for 1 h to ensure adsorption–desorption equilibrium prior to photocatalytic degradation. Photodegradation experiments lasted 160 min with 1-mL samples removed periodically and were carried out in an open quartz



**Figure 2.** Anatase TiO<sub>2</sub> NT arrays by anodization: (a) XRD pattern and (b) SEM image. As-prepared CuInSe<sub>2</sub> NCs: (c) XRD patterns and (d) TEM image. The inset of panel d is the corresponding selected area electrical diffraction (SAED) patterns of the as-prepared CuInSe<sub>2</sub> NCs.

photoreaction vessel with rapid stirring at 23 °C. The concentration of the residual MO was measured by a UV–vis spectrometer at 463 nm on the basis of the Beer–Lambert Law. The degradation efficiency of the MO can be defined as follows:

$$C_t / C_0 = A_t / A_0 \tag{1}$$

where  $C_0$  and  $C_t$  are the MO concentration at time zero and t (min), respectively, and  $A_0$  and  $A_t$  are the corresponding measured absorbance at time zero and time t, respectively. The total organic carbon (TOC) during the degradation process was monitored by a Shimadzu 500 TOC analyzer.

# 3. RESULTS AND DISCUSSION

Anodized TiO<sub>2</sub> NTs are usually amorphous in structure and need be crystallized by high temperature sintering before various applications. Figure 2a shows the XRD patterns of the as-obtained TiO2 NT arrays on the Ti sheet substrate after thermal annealing at 450  $^{\circ}C$  for 2 h. The TiO<sub>2</sub> NT arrays had a highly crystalline structure, judging from their sharp diffraction peaks and strong intensities. All of the diffraction peaks can be indexed to the anatase phase of TiO2 (JCPDS No. 21-1272), except for a few characteristic peaks that belong to the titanium substrate. For instance, four peaks located at  $2\theta = 25.3^{\circ}$ ,  $37.8^{\circ}$ ,  $48.05^{\circ}$ , and  $55.06^{\circ}$  could be attributed to the anatase (101), (004), (200), and (211) diffractions, respectively.<sup>41</sup> Figure 2b shows the FESEM image of the TiO<sub>2</sub> NT arrays. It reveals that the TiO<sub>2</sub> NT arrays consisted of hundreds of self-organized TiO2 NTs that grew vertically from the Ti substrate with thickness up to ~19.2  $\mu$ m. Meanwhile, it can be seen that these TiO<sub>2</sub> NTs were highly ordered with a diameter of  $\sim$ 110 nm. These results indicate that highly ordered anatase TiO<sub>2</sub> NT arrays were successfully fabricated by the anodization process.

Figure 2c and d shows the XRD patterns and a HRTEM image, respectively, of the as-synthesized CuInSe<sub>2</sub> NCs. XRD

patterns suggest the synthesized products have a highly crystalline nature, as shown in Figure 2c. All of the characteristic diffraction peaks of CuInSe<sub>2</sub> are clearly observed, such as (112), (220)/(204), and (116)/(312) peaks at  $2\theta$  = 26.6°, 44.7°, and 52.8°, respectively. Moreover, XRD results also indicate that the as-synthesized CuInSe<sub>2</sub> NCs have a tetragonal structure, and the calculated lattice parameters are as follows: a = b = 0.577 nm and c = 1.162 nm, which are in line with reported parameters of a = b = 0.578 nm and c = 1.162nm.<sup>42</sup> By the Scherrer equation, the average diameter of the CuInSe<sub>2</sub> NCs calculated from (112) diffraction peaks is about 13.5 nm. Figure 2d shows the HRTEM images of the CuInSe<sub>2</sub> NCs synthesized at 220 °C for 4 h, and the outline figure of those individual CuInSe2 NCs can be easily observed. The average size of the CuInSe<sub>2</sub> NCs is about 15 nm, which is a little bigger than the calculated average size from XRD results (~13.5 nm). As shown in Figure 2d, the lattice fringe d = 0.34nm and 0.2 nm can be assigned to the (112) and (220)/(204)lattice planes of the tetragonal CuInSe<sub>2</sub> NCs, respectively. Moreover, corresponding selected area electrical diffraction (SAED) patterns also indicate the nanocrystal structure of the as-synthesized CuInSe<sub>2</sub> NCs, as shown in the inset of Figure 2d; the three rings refer to the (112), (220)/(214), and (116)/(214)(312) reflection direction. It can be concluded that well crystallized tetragonal CuInSe<sub>2</sub> NCs were prepared, which were later used as a narrow-band semiconductor materials to decorate the wide-band semiconductor TiO<sub>2</sub> NT arrays.

Figure 3 shows the top-view SEM images of the highly ordered anatase  $TiO_2$  NT arrays before and after decoration with the as-synthesized CuInSe<sub>2</sub> NCs through an electrophoretic deposition process. Before deposition, the open tube mouth of  $TiO_2$  NT arrays can be clearly observed and the self-organized  $TiO_2$  NTs are highly ordered and closely packed

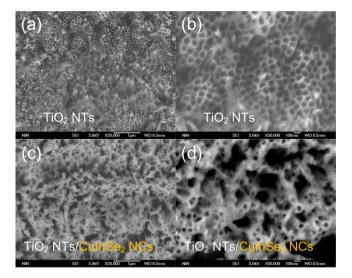


Figure 3. SEM images of the  $TiO_2$  NT arrays (a, b) before and (c, d) after decoration with the CuInSe<sub>2</sub> NCs (electrophoretic deposition at 150 V for 25 min).

together, as shown in Figure 3a and b. The average tube diameter is about 110 nm, with an average wall thickness of 15 nm. Figure 3c and d shows the morphologies of the TiO<sub>2</sub> NT arrays after decoration with CuInSe<sub>2</sub> NCs. By comparing with Figure 3a and b, it can be seen that the CuInSe<sub>2</sub> NCs were effectively deposited on the top surface (partially inside) of the TiO<sub>2</sub> NT arrays by the electrophoresis process at 150 V for 25 min. Moreover, the tube-pore mouths are still open after the deposition, indicating that well-ordered pore structure can be maintained, which is very important for photocatalytic performance.<sup>43</sup> These results suggest that the as-synthesized CuInSe<sub>2</sub> NCs were successfully immobilized with TiO<sub>2</sub> NT arrays, the deposition process does not damage the highly ordered TiO<sub>2</sub> NT arrays, and the heterogeneous structure of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs was obtained.

To further investigate the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT heterostructured films, an elemental analysis was carried out using energy-dispersive X-ray spectroscopy (EDX). Supplementary Figure S1 shows a typical EDX spectrum of the CuInSe<sub>2</sub> NCs/ TiO<sub>2</sub> NT film after 25 min electrophoretic deposition, and the characteristic peaks belonging to Ti, Cu, In, Se were observed, confirming the successful deposition of CuInSe<sub>2</sub> NCs. It should be noted that the EDX results show elemental composition of only the top surface of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT array (within several micrometers at most). The content ratio of Cu, In, and Se is about 2.4 atom %, 2.7 atom %, and 7.8 atom %, respectively, as shown in Table 1. The element ratio of Cu:In:Se roughly agrees with the formula ratio 1:1:2 of CuInSe<sub>2</sub>. Supplementary Figure S2 shows that different elements of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT films vary with increased electrophoresis time. Cu, In, and Se contents increased from 2.4 to 12.0 atom %, from 2.7 to 19.9 atom %, and from 7.8 to 47.5 atom %, respectively, with electrophoresis duration increasing from 25 to 75 min. The elemental content of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films were summarized as shown in Table 1, and it can be easily found that the ratio of Cu/Ti, In/Ti, and Se/Ti increased with the electrophoresis time. Meanwhile, an increase in the relative concentration of In and Se can be observed with the increase of deposition time, which may be because the CuInSe<sub>2</sub> NCs have a relatively large size distribution, so during the electrophoresis procedure smaller (more mobile) CuInSe<sub>2</sub> NCs get deposited onto the TiO<sub>2</sub> NTs first, followed by the bigger ones. Smaller and bigger CuInSe<sub>2</sub> NCs do not have the same composition, causing the gradual change in the total composition of the CuInSe<sub>2</sub> NCs layer. The results indicate that an increasing amount of CuInSe<sub>2</sub> NCs was successfully decorated on the surfaces of the TiO<sub>2</sub> NT arrays as the deposition duration increased.

UV-vis absorption spectrum can be used to investigate optical properties of the  $CuInSe_2 NCs/TiO_2 NTs$  films. Figure 4 shows the normalized UV-vis absorption spectra of the pure

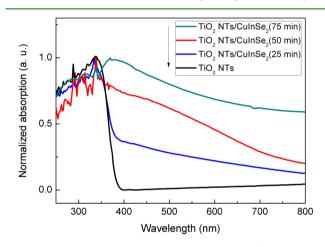


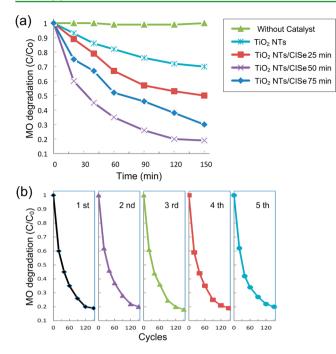
Figure 4. Normalized UV–vis absorption spectra of the  $CuInSe_2$  NCs/TiO<sub>2</sub> NT arrays films with electrophoretic time increased from 25 min to 75 min.

 $TiO_2$  NT film and the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films with different deposition duration from 25 min to 75 min. It can be seen that the absorption range of the undecorated  $TiO_2$  NT film lies only within the ultraviolet region (<380 nm), while that of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT films cover almost the entire visible region (about 400–700 nm). Meanwhile, an increased absorption was also observed as the CuInSe<sub>2</sub> NCs deposition duration increased.

Photocatalytic activities of the  $CuInSe_2 NCs/TiO_2 NT$  films were evaluated through degradation of MO aqueous solution, with degradation kinetics shown in Figure 5a. It can be seen

Table 1. Element Contents Based on EDX Results and First Order Constant k Values for the CuInSe<sub>2</sub> NC/TiO<sub>2</sub> NT Films with Various Electrophoresis Times

	EDX results, atom %					
samples	Cu (error)	In (error)	Se (error)	Ti (error)	O (error)	av $k$ , min <sup>-1</sup>
CuInSe <sub>2</sub> NCs/TiO <sub>2</sub> NTs; 75 min	12.0 (2.17)	19.9 (0.67)	47.5 (0.38)	7.79 (0.46)	12.9 (0.33)	0.006
CuInSe <sub>2</sub> NCs/TiO <sub>2</sub> NTs; 50 min	7.3 (2.31)	10.3 (0.67)	26.0 (0.4)	37.5 (0.46)	18.9 (0.44)	0.016
CuInSe <sub>2</sub> NCs/TiO <sub>2</sub> NTs; 25 min TiO <sub>2</sub> NTs	2.4 (3.22)	2.7 (0.85)	7.8 (0.55)	46.8 (0.59)	40.3 (0.73)	0.010 0.003



**Figure 5.** (a) Photocatalytic degradation kinetics of MO aqueous solution with the CuInSe<sub>2</sub> NC/TiO<sub>2</sub> NT arrays films as photocatalytic panel. (b) Photocatalytic cycling test using the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT arrays with 50 min electrophoretic deposition.

that, for the blank testing, negligible MO degradation was observed when no photocatalyst applied, which indicates the MO solution was stable as a probe to test the activities of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films. When a photocatalyst panel was included, significant MO degradation was achieved immediately for all of the samples. Figure 5a shows that around 28% of the MO was photo-catalytically degraded by using the pure anatase TiO<sub>2</sub> NT array film after 150 min of irradiation, while over 50% of the MO was degraded using the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films, indicating an enhanced photocatalytic activity. Supplementary Figure S3 shows the curves of  $-\ln(C/C_0)$  variation with degradation time, and a liner relation can be easily observed for all of the samples, which indicates that the photocatalytic degradation of the MO followed pseudo-firstorder kinetics and the kinetic reaction can be expressed as follows:

$$A_t = A_0 e^{-kt} \tag{2}$$

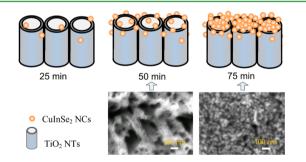
where  $k \pmod{1}$  is the degradation rate constant. The calculated first order constant k values for all of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films are summarized in Table 1. It can be seen that the degradation constant k value of the pure TiO<sub>2</sub> NTs is only about 0.003 min<sup>-1</sup>, which increased two-fold to 0.006 min<sup>-1</sup> for the  $\mbox{CuInSe}_2\ \mbox{NCs}/\mbox{TiO}_2\ \mbox{NTs}$  film with 25 min deposition. When the CuInSe<sub>2</sub> NCs deposition duration increased from 25 to 50 min, the degradation rate constant k value increased from 0.006 min<sup>-1</sup> to the maximum value of 0.016 min<sup>-1</sup>. However, a decreased k value was observed (0.010  $min^{-1}$ ) when the CuInSe<sub>2</sub> NCs deposition duration was 75 min. Therefore, 50 min electrophoretic deposition is the optimized duration in the study. The relationship between the degradation rate constant kvalues and the CuInSe<sub>2</sub> NCs electrophoretic deposition time is shown in Supplementary Figure S4. Although the TOC testing showed a hysteretic degradation kinetic (see Supplementary Figure S5), the MO solution was found evidently decompounded by the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs films. On the basis of the above results, it can be concluded that the photocatalytic performance of the TiO<sub>2</sub> NT array film can be greatly enhanced by introducing the CuInSe<sub>2</sub> NCs. The enhanced photocatalytic activity is mainly ascribed to two aspects:<sup>44,45</sup>

(a) The CuInSe<sub>2</sub> NCs improve the absorption efficiency of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT array films almost through the entire visible-light region, as shown in Figure 4. This allows more photons to be absorbed and generates more active electron-hole pairs, which subsequently contributes to the photocatalytic enhancement;

(b) The formation of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT heterojunctions can suppress the recombination of the photogenerated electron-hole pairs and prolong the charge-carrier's lifetime, which also benefited the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT film's photocatalytic performance. The band gap structures and their Fermi energy levels are schematically shown in Figure 1b.

Moreover, BET analysis indicates the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs-50 min sample has a specific area of ~39 m<sup>2</sup>/g, and cycling test results indicate that the as-obtained CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs heterojunction films are very stable and have a repeatable performance. Figure 5b shows that the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs-50 min sample kept its high activity after 5 cycles. This result indicates the CuInSe<sub>2</sub> NCs were firmly attached to the TiO<sub>2</sub> NT arrays and thus successfully immobilized.

To better understand the influence of  $CuInSe_2 NCs$  on photocatalytic activities of the  $CuInSe_2 NCs/TiO_2 NT$  heterojunction films, we propose a possible deposition mechanism of the  $CuInSe_2 NCs$  on the  $TiO_2 NT$  array films with the electrophoresis time as illustrated in Figure 6. It can be



**Figure 6.** Schematic of the deposition process of the CuInSe<sub>2</sub> NCs on the  $TiO_2$  NTs array films with the electrophoresis time, the insets are the corresponding SEM images.

observed from Figure 6 that, at 25 min, scattered  $CuInSe_2$ nanocrystal particles were deposited on the TiO<sub>2</sub> NTs forming a CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs heterojunction structure. The CuInSe<sub>2</sub> NCs could be activated by visible-light photons, and the photogenerated electrons would migrate to the TiO<sub>2</sub> NTs enabling the visible-light photocatalytic activity with an overall enhancement as we observed in Figure 5. By increasing the electrophoresis time to 50 min, more CuInSe<sub>2</sub> NCs were decorated onto the TiO<sub>2</sub> NTs and acted as a visible-light sensitizer for photocatalytic reaction, without blocking the open tube mouth of the TiO<sub>2</sub> NTs at the same time. The CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs-50 min sample shows the highest degradation rate ( $k = 0.016 \text{ min}^{-1}$ ), which was almost 5 times higher than that of the pure TiO<sub>2</sub> NT arrays ( $k = 0.003 \text{ min}^{-1}$ ), as shown in Table 1 and Supplementary Figure S4. At 75 min deposition

time, however, excessive CuInSe<sub>2</sub> NCs were deposited on the TiO<sub>2</sub> NTs. The inset SEM of Figure 6 shows that the CuInSe<sub>2</sub> NCs started forming a film-like layer and blocked most of the open tube mouths of the TiO2 NT arrays. With fewer open tube mouths, the molecular transportations channels are reduced: both MO molecules and photo-generated ROS diffusion will be restricted out of and into solution, leading to the decrease in degradation rate, k, back to 0.010 min<sup>-1</sup>. Therefore, there should be a balance between CuInSe<sub>2</sub> NCs loading mass and retaining the open tube-mouth configuration for achieving the best photocatalytic activity of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs heterojunction films. Since the above assumption is consistent with SEM, EDX, and photocatalytic degradation results, we believe that the CuInSe, NCs with a proper electrophoretic deposition quantity play a significant role in the enhanced visible-light photocatalytic activities of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT heterojunction films.

## 4. CONCLUSIONS

Highly-ordered TiO<sub>2</sub> NT arrays and well-dispersed tetragonal CuInSe<sub>2</sub> NCs were successfully prepared by using a facile anodization method and a solvothermal method, respectively. The highly ordered TiO<sub>2</sub> NT arrays with open tube mouths were decorated with the CuInSe<sub>2</sub> NCs through an electrophoretic process, forming a CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NTs heterojunction film. Compared with pure TiO2 NTs, the asobtained CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT films show enhanced photocatalytic activity towards degrading MO. The enhanced photocatalytic activities were attributed to the formation of CuInSe<sub>2</sub>/TiO<sub>2</sub> heterojunctions, which improve the visible-light absorption efficiency and inhibit the recombination of the photogenerated electron-hole pairs. SEM, EDX, and UV-vis results indicate that increased CuInSe<sub>2</sub> NCs were immobilized on the TiO<sub>2</sub> NTs films with increasing electrophoresis time. The optimized electrophoretic time is 50 min in this study, and the photocatalytic activity of the CuInSe<sub>2</sub> NCs/TiO<sub>2</sub> NT film was enhanced more than 5 times compared to that of undecorated TiO<sub>2</sub> NT films, due to a proper CuInSe<sub>2</sub> NCs loading and the well-maintained open tube-mouth configuration. These results are expected to be helpful for further understanding of the preparation and photocatalytic properties of CuInSe<sub>2</sub> nanocrystal decorated TiO<sub>2</sub> nanotube arrays film photocatalysts.

#### ASSOCIATED CONTENT

#### **S** Supporting Information

EDX spectrum in detail and additional photocatalytic results and analysis. This material is available free of charge via the Internet at http://pubs.acs.org.

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## Notes

The authors declare no competing financial interest.

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#### REFERENCES

(1) Qu, X.; Brame, J.; Li, Q.; Alvarez, P. J. J. Acc. Chem. Res. 2012, 46, 834-843.

(2) Lim, Y. W. L.; Tang, Y.; Cheng, Y. H.; Chen, Z. Nanoscale 2010, 2, 2751–2757.

(3) Choudhury, B.; Borah, B.; Choudhury, A. Photochem. Photobiol. 2012, 88, 257–264.

(4) Dholam, R.; Patel, N.; Adami, M.; Miotello, A. Int. J. Hydrogen Energy 2009, 34, 5337-5346.

(5) Brame, J. A.; Hong, S. W.; Lee, J.; Lee, S.-H.; Alvarez, P. J. J. *Chemosphere* **2013**, *90*, 2315–2319.

(6) Jiang, Z.; Tang, Y.; Tay, Q.; Zhang, Y.; Malyi, O. I.; Wang, D.; Deng, J.; Lai, Y.; Zhou, H.; Chen, X.; Dong, Z.; Chen, Z. *Adv. Energy Mater.* **2013**, *3*, 1368–1380.

(7) Fischer, B. B.; Krieger-Liszkay, A.; Eggen, R. I. L. Environ. Sci. Technol. 2004, 38, 6307-6313.

- (8) Khan, M. A.; Akhtar, M. S.; Woo, S. I.; Yang, O. B. Catal. Commun. 2008, 10, 1–5.
- (9) Macak, J. M.; Zlamal, M.; Krysa, J.; Schmuki, P. Small 2007, 3, 300–304.
- (10) Choudhury, B.; Choudhury, A. Curr. Appl. Phys. 2013, 13, 1025–1031.
- (11) Foong, T. R. B.; Chan, K. L.; Hu, X. Nanoscale 2012, 4, 478–485.
- (12) Foong, T. R. B.; Shen, Y.; Hu, X.; Sellinger, A. Adv. Funct. Mater. 2010, 20, 1390–1396.
- (13) Jha, P. K.; Gupta, S. K.; Lukačević, I. Solid State Sci. 2013, 22, 8–15.
- (14) Mankad, V.; Gupta, S. K.; Jha, P. K. Phys. E 2011, 44, 614-617.

(15) Mishra, S.; Jha, P.; Pratap, A. J. Therm. Anal. Calorim. **2012**, 107, 65–68.

(16) Wang, P.; Tang, Y.; Dong, Z.; Chen, Z.; Lim, T.-T. J. Mater. Chem. A 2013, 1, 4718-4727.

(17) Wang, S.; Qian, H.; Hu, Y.; Dai, W.; Zhong, Y.; Chen, J.; Hu, X. Dalton Trans. 2013, 42, 1122-1128.

(18) Xiu, Z.-m.; Zhang, Q.-b.; Puppala, H. L.; Colvin, V. L.; Alvarez, P. J. J. Nano Lett. **2012**, *12*, 4271–4275.

(19) Mohapatra, S. K.; Misra, M.; Mahajan, V. K.; Raja, K. S. J. Phys. Chem. C 2007, 111, 8677–8685.

(20) Park, J. H.; Kim, S.; Bard, A. J. Nano Lett. 2006, 6, 24-28.

- (21) Woan, K.; Pyrgiotakis, G.; Sigmund, W. Adv. Mater. 2009, 21,
- 2233-2239.
- (22) Zheng, Q.; Zhou, B.; Bai, J.; Li, L.; Jin, Z.; Zhang, J.; Li, J.; Liu, Y.; Cai, W.; Zhu, X. *Adv. Mater.* **2008**, *20*, 1044–1049.
- (23) Liao, Y.; Brame, J.; Que, W.; Xiu, Z.; Xie, H.; Li, Q.; Fabian, M.; Alvarez, P. J. J. Hazard. Mater. **2013**, 260, 434–441.
- (24) Li, L.; Liu, X.; Zhang, Y.; Nuhfer, N. T.; Barmak, K.; Salvador, P. A.; Rohrer, G. S. ACS Appl. Mater. Interfaces **2013**, *5*, 5064–5071.

(25) Li, L.; Liu, X.; Zhang, Y.; Salvador, P. A.; Rohrer, G. S. Int. J. Hydrogen Energy 2013, 38, 6948-6959.

- (26) Daskalaki, V. M.; Antoniadou, M.; Li Puma, G.; Kondarides, D. I.; Lianos, P. *Environ. Sci. Technol.* **2010**, *44*, 7200–7205.
- (27) Qian, S.; Wang, C.; Liu, W.; Zhu, Y.; Yao, W.; Lu, X. J. Mater. Chem. 2011, 21, 4945–4952.

(28) Meng, Z.-D.; Zhu, L.; Ye, S.; Sun, Q.; Ullah, K.; Cho, K.-Y.; Oh, W.-C. Nanoscale Res. Lett. 2013, 8, 1–10.

- (29) Lo, S.-C.; Lin, C.-F.; Wu, C.-H.; Hsieh, P.-H. J. Hazard. Mater. 2004, 114, 183-190.
- (30) Li, G.; Wu, L.; Li, F.; Xu, P.; Zhang, D.; Li, H. Nanoscale 2013, 5, 2118–2125.
- (31) Zhang, K.; Guo, L. Catal. Sci. Technol. 2013, 3, 1672-1690.

(32) Koo, B.; Patel, R. N.; Korgel, B. A. J. Am. Chem. Soc. 2009, 131, 3134–3135.

(33) Panthani, M. G.; Akhavan, V.; Goodfellow, B.; Schmidtke, J. P.; Dunn, L.; Dodabalapur, A.; Barbara, P. F.; Korgel, B. A. *J. Am. Chem. Soc.* **2008**, *130*, 16770–16777.

(34) Aldakov, D.; Lefrancois, A.; Reiss, P. J. Mater. Chem. C 2013, 1, 3756–3776.

(35) Grimes, C. A. J. Mater. Chem. 2007, 17, 1451-1457.

(36) Mor, G. K.; Kim, S.; Paulose, M.; Varghese, O. K.; Shankar, K.; Basham, J.; Grimes, C. A. *Nano Lett.* **2009**, *9*, 4250–4257.

(37) Prakasam, H. E.; Shankar, K.; Paulose, M.; Varghese, O. K.; Grimes, C. A. J. Phys. Chem. C 2007, 111, 7235–7241.

(38) Zhang, J.; Que, W.; Shen, F.; Liao, Y. Sol. Energy Mater. Sol. Cells **2012**, 103, 30-34.

(39) Shen, F.; Que, W.; Liao, Y.; Yin, X. Ind. Eng. Chem. Res. 2011, 50, 9131–9137.

(40) Miyasaka, T.; Kijitori, Y. J. Electrochem. Soc. 2004, 151, 1767–1773.

(41) Allam, Nageh K.; G., C. A. J. Phys. Chem. C Lett. 2009, 113, 7996–7999.

(42) Guo, Q.; Kim, S. J.; Kar, M.; Shafarman, W. N.; Birkmire, R. W.; Stach, E. A.; Agrawal, R.; Hillhouse, H. W. Nano Lett. **2008**, *8*, 2982–

2987. (43) Mazzarolo, A.; Lee, K.; Vicenzo, A.; Schmuki, P. Electrochem. Commun. **2012**, 22, 162–165.

(44) Shen, F.; Que, W.; He, Y.; Yuan, Y.; Yin, X.; Wang, G. ACS Appl. Mater. Interfaces **2012**, *4*, 4087–4092.

(45) He, Z.; Que, W.; Chen, J.; Yin, X.; He, Y.; Ren, J. ACS Appl. Mater. Interfaces 2012, 4, 6816–6826.